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Abstract
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Keywords
potassium compounds, surface structure, charge density waves, scanning tunnelling microscopy, crystal structure

Comments
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Atomic structure and charge-density waves of blue bronze \( \text{K}_{0.3}\text{MoO}_3 \) (20\( \bar{1} \)) by variable-temperature scanning tunneling microscopy

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Blue bronze \( \text{K}_{0.3}\text{MoO}_3 \) has been the focus of a number of scattering, transport, scanning tunneling microscopy (STM), and theoretical studies that have provided insight into the relation between atomic structure and charge-density wave (CDW) formation. However, the full extent of a relation of the CDWs to the atomic lattice and the microscopic origin of CDW pinning are still not completely resolved. In this study STM is used to distinguish the atomic structure and CDWs at the (20\( \bar{1} \)) surface. Within the STM’s spatial resolution, the CDWs are incommensurate with the lattice at midrange temperatures and approach commensurability at low temperatures. Incommensurate CDWs are present on the surface and the degree of the incommensurability between blue bronze lattice and CDW lattice agree well with those determined from bulk scattering techniques.

Complex oxides continue to receive extensive attention due to their rich physical properties arising from electronegativity interactions, charge and spin ordering, etc. Classic among this class of compounds are the low-dimensional molybdates and tungstates that exhibit charge-density wave (CDW) instabilities.1 The “blue bronzes” \( \text{A}_{0.3}\text{MoO}_3 \), where \( \text{A}=\text{K}, \text{Rb}, \text{or} \text{TI} \) in particular have been the focus of a number of scattering,2–7 transport, scanning tunneling microscopy (STM),8 and theoretical studies that have provided insight on the relation of atomic and CDW structure and concomitant phase transitions.1 However several fundamental issues remain. For example, the precise relation between the CDWs and the atomic lattice and the microscopic origin of CDW pinning in these materials are unresolved.

Almost from its invention, STM has been applied to CDW compounds.8–15 Since STM images the surface density of states near the Fermi level, it can distinguish atomic structure at a surface from spatial variations of charge density when they have a period different from that of the atomic lattice. Attempts to image blue bronzes at room temperature have yielded high spatial resolution.16–18 Walter et al. found no evidence for CDWs at the surface below the Peierls transition temperature \( T_P=180 \) K.8 Recently, Brun et al.19,20 examined the surface of \( \text{Rb}_{0.3}\text{MoO}_3 \) in the temperatures in the range of 63–78 K with STM interpreting the image contrast as superposition of surface topography and CDW. Here we use STM imaging of \( \text{K}_{0.3}\text{MoO}_3 \) to characterize the atomic structure of the surface. We observe CDW modulations that are incommensurate with the lattice at midrange temperatures below Peierls transition and that become indistinguishable from the lattice at lower temperatures near 100 K. We relate these results to the crystallographic structure of \( \text{K}_{0.3}\text{MoO}_3 \).

Blue bronze (in this case \( \text{K}_{0.3}\text{MoO}_3 \)) is monoclinic with space group \( \text{C}2/m \) and lattice parameters \( a=1.825 \text{ nm}, b=0.756 \text{ nm}, c=0.9885 \text{ nm}, \beta=117.5^\circ \).7,21 The structure consists of slabs of distorted \( \text{MoO}_6 \) octahedra separated by and bonded through K atoms. This layered structure cleaves easily along the (20\( \bar{1} \)) plane. In the cleavage plane the octahedra form corner-shared chains along the \( \text{b} \) direction with additional edge-shared octahedral located 0.06 nm lower at one-half of the junctions. Figure 1 illustrates that the (20\( \bar{1} \)) surface plane contains three different Mo environments. The electron density is higher at the Mo II and Mo III than at the Mo I because the average valence of the former is 5.6, while that of Mo I is 5.75.22 Consequently, Mo II and Mo III contribute significantly to the density of states at the Fermi level, whereas Mo I does not. At room temperature blue bronze \( \text{K}_{0.3}\text{MoO}_3 \) is a quasi-one-dimensional conductor that undergoes a Peierls transition at \( T_P=180 \) K that fully gaps the Fermi surface, resulting in semiconducting behavior well below \( T_P \). Neutron and x-ray scattering have shown that the CDWs are incommensurate with the atomic lattice below 180 K,4 with a CDW wave vector \( q_{\text{CDW}} \) component along the

FIG. 1. (Color online) The structure of the \( \text{K}_{0.3}\text{MoO}_3 \) (20\( \bar{1} \)) surface illustrating three different environments of Mo atoms (white circles are K, gray circles are Mo, small dark gray circles are O; note that K, Mo, and O are drawn not to scale). Mo octahedral are shaded with violet (light gray) color. There are two representations of the surface structure present: (a) traditional (\( \text{b}; \text{a}+2\text{c} \)) and (b) conventional crystallographic (\( \text{b}; \text{112} \)).

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highly conducting $b^*$ direction of $\sim 0.26$ $b^*$. Grazing incidence x-ray diffraction measurements did not reveal any dependence of the lattice parameter on the depth probed, suggesting that the surface and bulk structures should agree.

Single crystals of blue bronze (K$_{0.3}$MoO$_3$) were grown in quartz ampoules in the presence of an Ar atmosphere. A mixture of MoO$_3$, MoO$_2$, K$_2$MoO$_4$ powders with initial mass ratio 8.04 g: 1.53 g: 2.86 g was hot-pressed into a pellet. The pellet was placed into a quartz ampoule 21.5 cm long and degassed under vacuum, and then the ampoule was back-filled with Ar gas and sealed. The crystals were grown for 10 days at 860 K. The phase composition of the growth was determined by x-ray diffraction.

STM images were obtained under UHV conditions ($P<10^{-9}$ mBar). The sample was mounted on a degenerately doped silicon base with UHV epoxy (Epotec, H20D) and cleaved under vacuum with a homebuilt attachment. STM measurements were performed with an Omicron VT-25 AFM/STM equipped with a KONTY cryostat. The sample was cooled using liquid N$_2$, and its temperature determined to be 10$^{-6}$ mBar. Filled state images were acquired from the surface. In region 3, the unit-cell-size corrugations have the same lateral dimensions but amplitudes of $\sim 0.02$ nm which we attribute to tunneling with a W atom at the tip apex. In region 2, corrugations have the same lateral dimensions but amplitudes of $\sim 0.06$ nm. Since the DOS of Mo is higher than that of W in the voltage range 0–2 V (Ref. 24) this larger corrugation amplitude suggests the presence of a Mo atom at the tip (the interpretation has been done assuming that DOS of the atom at the tip apex is similar to the DOS of the bulk metal). In region 1, the contrast is completely reversed. This is consistent with an oxygen ter-
FIG. 4. Variable temperature STM on a vacuum cleaved surface of blue bronze: (a) 300 K, (b) 140 K, and (c) 100 K. FFT of each image is presented on the right-hand side of each one. Splitting the FFT peaks is obvious in image (e). This could be attributed to the formation of incommensurate CDW on the crystal surface. Lowering the temperature (to 100 K) results in structure change shown in the figure at the bottom. Sample voltages: (a) 300 K, +2 V; current, 0.2 nA (W tip); (b) 140 K, sample voltage, +1.5 V; current, 0.2 nA (W tip); (c) 100 K, sample voltage, +1.5 V; current, 0.3 nA (W tip). The line profile in the b direction from 4(b) is reproduced in Fig. 5(a).

minimized tip, which should have a stronger interaction with the K sites than with the oxide octahedra. Elucidation of the detailed structures of the tip atom clusters requires first principles calculations, but the above considerations allow the stable contrast of region 1 to be associated with the atomic structure as shown in Fig. 2.

For the variable temperature constant current STM measurements in Fig. 4, the “apparent” surface structure changes as the temperature is lowered from 300 K to 100 K. The “apparent” unit cell dimensions determined from real space images and fast Fourier transforms (FFT) of the image contrast are listed in Table I. At 300 K only one periodic structure is present on the (201) surface, with lattice vectors \( \mathbf{I} = 1.2 \text{ nm} \) and \( \mathbf{b} = 0.76 \text{ nm} \). The modulus of the lattice vector matches the known bulk distance between (112) planes, and the angle between vectors \( \mathbf{I} \) and \( \mathbf{b} \) also agree with the bulk angle. At 140 K (below the Peierls transition), two periodic structures are evident from peak splitting in the FFT, Fig. 4(b). One structure has the lattice parameters of the atomic structure at 300 K, consistent with the small change in bulk lattice parameters (<1 %) in this temperature range. The second structure is smaller (vector \( \mathbf{I} = 1.2 \text{ nm} \), vector \( \mathbf{b} = 0.71 \text{ nm} \)) with slightly different angles between the surface unit vectors (see Table I). We attribute this structure to the CDWs on the surface based on two equivalent analytical procedures elaborated on here. The CDWs can be quantified from STM image data in two ways. First, a profile is extracted from the image along the b direction, as shown in Fig. 5(a). The corrugation associated with the CDW is not distinguished in the profile, but the effect of the additional intensity at the atomic sites is clear. The superposition of the two (CDW’ and atomic sites’

<table>
<thead>
<tr>
<th>Temperature</th>
<th>Vector I</th>
<th>Vector b</th>
<th>Angle between I and b</th>
</tr>
</thead>
<tbody>
<tr>
<td>300 K</td>
<td>1.2 nm</td>
<td>0.76 nm</td>
<td>110°</td>
</tr>
<tr>
<td>140 K</td>
<td>1.2 nm</td>
<td>0.76 nm</td>
<td>110°</td>
</tr>
<tr>
<td>100 K</td>
<td>1.0 nm</td>
<td>0.76 nm</td>
<td>92°</td>
</tr>
</tbody>
</table>

TABLE I. The change of surface unit cell dimensions with temperature.
height corrugations) results in an underlying low-frequency component with a period 11 times that of the lattice. This represents a lattice mismatch of about 9%, which agrees with bulk measurements [−9% at 140 K (Ref. 14)]. In an alternative approach, the components of the FFT can be separated, reverse transformed, and superimposed, as shown in Fig. 5(b). Here the separation of the two lattices is evident and illustrates mismatch on the order of 10%, along the b direction and almost commensurate alignment along the a + b + 2c direction.

On decreasing the temperature to 100 K the additional peaks in the FFT of the STM image are no longer resolved [Figs. 4(c) and 4(f)]. Either the CDWs become commensurate with the atomic lattice or the difference between the two is so small as to be undetectable with the technique.

Previous authors have imaged charge density waves in a similar compound, Rb$_{0.3}$MoO$_3$.15,16 In that case the CDW structure was spatially close to the lattice dimensions that the superposition of the two structures was apparent in superlattice spots in the FFT. In the present case the incommensurate CDW are imaged directly. We suggest that the superlattice spots are not as apparent in our study due to the higher temperature at which the CDWs were imaged. Alternatively, a recent analysis by Canadell and collaborators25 of a similar compound Rb$_{0.3}$MoO$_3$ indicates that a sensitivity of a detection of alkali-metal atoms may play a role in the interpretation of images.

In summary, the CDW phase transition was observed on the K$_{0.3}$MoO$_3$ surface with in situ variable temperature STM. The atomic structure at the surface exhibited the lattice parameters consistent with those of the bulk at all temperatures. Incommensurate CDWs were present on the surface and the degree of incommensurability with the atomic lattice was determined and found to be in a good agreement with those determined from bulk scattering techniques.

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